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Fast Tuning of Double Fano Resonance Using A Phase-Change Metamaterial Under Low Power Intensity

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In this work, we numerically demonstrate an all-optical tunable Fano resonance in a fishnet metamaterial(MM) based on a metal/phase-change material(PCM)/metal multilayer. We show that the displacement of the elliptical nanoholes from their centers can split the single Fano resonance (FR) into a double FR, exhibiting higher quality factors. The tri-layer fishnet MMs with broken symmetry accomplishes a wide tuning range in the mid-infrared(M-IR) regime by switching between the amorphous and crystalline states of the PCM ($\text{Ge}_2\text{Sb}_2\text{Te}_5$). A photothermal model is used to study the temporal variation of the temperature of the $\text{Ge}_2\text{Sb}_2\text{Te}_5$ film to show the potential for switching the phase of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ by optical heating. Generation of the tunable double FR in this asymmetric structure presents clear advantages as it possesses a fast tuning time of 0.36 ns, a low pump light intensity of $9.6 \mu\text{W}/\mu\text{m}^2$, and a large tunable wavelength range between 2124 nm and 3028 nm. The optically fast tuning of double FRs using phase change metamaterials(PCMMs) may have potential applications in active multiple-wavelength nanodevices in the M-IR region.

Recently, Fano resonance(FR) excited by an interference coupling between bright(radiating) modes and dark (non radiating) modes in metallic plasmonic nanostructures has been widely investigated^{1,2}. It has generated much interest because of its application in the fields of chemical and biological sensing^{3,4}, surface enhanced raman spectroscopy⁵, active plasmonic switching⁶, and slow light devices⁷. FR exhibits an asymmetric line shape with a sharp feature in transmission, reflection or absorption spectrum over an extremely narrow frequency range^{8,9}. There have been a few attempts to create FRs in metallic plasmonic nanostructures, such as core-shell structure¹⁰, metallic nanowire arrays^{11,12}, composite cut-wire structures¹³ plasmonic dimmers/clusters¹⁴ and photonic metamaterials(MMs)^{15–18}. Among the various MMs, the configurations with broken symmetry are the most common approach for exciting narrow FRs. These asymmetric MMs give rise to a non-uniform electromagnetic environment around the resonators, leading to the effective coupling between bright modes and dark modes¹⁹.

The ability to control the central wavelength of a FR in MMs can find more flexible and critical applications in practice such as tunability of dispersion and group velocity of light¹⁸, multispectral imaging/detection²⁰, broadband delay lines²¹. However, most researchers focus on passive tuning of FRs by adjusting geometric parameters of the MMs^{22–26} which limits the scope of applications. Recently, active tuning of the optical response of MMs has attracted growing interest owing to its ultra-short modulation time and on-to-off switching capabilities^{27,28}. Therefore, for the purpose of actively controlling FRs, MMs integrated with optically active materials is a promising approach. Quite recently, a variety of approaches to modulate FRs in active MMs have emerged. Pryce et al exploited tunable FRs in MMs by means of mechanical deformation of an elastomeric substrate²⁹. Belotelov et al presented a magnetically tunable FR in a magneto-plasmonic MM³⁰. Kurter et al reported cryogenic temperature-dependent electromagnetically induced transparency(EIT) MM³¹. Samson et al demonstrated a frequency shift of a FR mode in a plasmonic planar MM by electrically tuning the dielectric properties of an adjacent chalcogenide glass layer³². Mousavi et al used the plasmonic response of graphene to tune a FR in the mid-infrared(M-IR) region³³. However, FRs in these MMs are magnetically and electrically tuned hence only responding on a scale of several microseconds to seconds³⁴. In addition, the integration of the required MM structures as well as the electrodes for the tuning of active dielectric layers can be difficult. Therefore, an appealing route that is notable for fast tunable FRs is to optically adjust the optical properties of the MMs. To this end, Gu et al presented active optical control of EIT MMs by integrating photoactive silicon(Si) islands into resonant



elements¹⁸. However, this MM works only in the terahertz (THz) regime, and its structural geometry is complicated to fabricate in the optical region, which limits its potential use. Zhu et al proposed a fast all-optical tunable FR in the optical frequency range, using a plasmonic MM made of gold and polycrystalline lithium niobate (LiNbO_3)⁸. However, this two dimensional (2D) planar MM system is based on a single metallic grating layer. It has been suggested that multiple layers of perforated metal–dielectric–metal (MDM) stacks with up to 100 to 200 layers along the propagation direction show a promising approach for three dimensional (3D) optical MMs^{35–37}. Such 3D MMs can possess a thickness much larger than the free space wavelength in the optical region and excite a strong magneto-inductive coupling between neighboring functional layers under a normally incident light. The tight coupling between adjacent LC resonators through mutual inductance results in a low loss³⁶. Thus 3D multilayer MMs are likely to be required for nearly all potential applications, such as cloaking and invisible materials^{35–38} as well as tunable metamaterials. Here we choose phase change material (PCM) as the dielectric interlayer, using which, we have previously shown designs for low loss tunable negative index material³⁹ and a polarization independent tunable perfect absorber⁴⁰. Thus, PCM based active tuning of FRs in three dimensional (3D) multilayer MMs is an interesting direction to pursue. To the best of our knowledge, it is still a formidable challenge to obtain a fast, low-power all optical tunable FR using 3D multilayer MMs in the optical regime.

Here, we numerically demonstrate that the resonant frequency of FRs can be optically tuned using PCMs in the mid-infrared (M-IR) region. Our structure is composed of an elliptical nanohole array (ENA) perforating through a Metal–Dielectric–Metal (MDM) trilayer, and a prototypical PCM: $\text{Ge}_2\text{Sb}_2\text{Te}_5$, is selected as the dielectric layer. A single FR is observed in the symmetric MDM–ENA. This FR response is caused by the interaction between the bright and dark modes. The elliptical nanohole resonator array shows electric resonance (namely localized surface plasmon) which excite the bright

modes, whereas MDM multilayers sustain a magnetic resonance (namely inductive-capacitive resonance) to induce the dark modes at the same frequency^{18,41}. As soon as the elliptical holes are displaced from their central positions, the single magnetic resonance would be divided into two magnetic resonances at different wavelengths. A double FR appears, owing to the destructive interference between the electric resonance and the two magnetic resonances at different wavelengths. This interference can enhance the FR. By using $\text{Ge}_2\text{Sb}_2\text{Te}_5$, a broad spectral tunability (from 2124 nm to 3082 nm) of the double FRs in the M-IR region can be obtained by switching between the amorphous and crystalline states of the PCMs⁴². Importantly, a heat model is constructed to examine the temporal variation of the temperature of the $\text{Ge}_2\text{Sb}_2\text{Te}_5$ layer in the asymmetric MDM–ENA. The model shows that the temperature of the amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ layer can be raised from room temperature to >433 K (phase transition point of $\text{Ge}_2\text{Sb}_2\text{Te}_5$)^{43,44} in just 0.36 ns, with a low operating pump light intensity of $9.6 \mu\text{W}/\mu\text{m}^2$. Therefore, our proposed structure can rapidly tune the double FR under a low light intensity in the optical region using $\text{Ge}_2\text{Sb}_2\text{Te}_5$. The optically active tuning of double FRs in these phase change metamaterials (PCMMs) may open up avenues for designing tunable multi-wavelength nanostructures in the optical regime such as slow light devices, ultrasensitive sensors and nonlinear devices. These PCMMs possess a simple geometry which remains compatible with standard photolithography patterning and can be easily experimentally realized in the optical region. Finally, it should be noted that PCMs do not require any energy to maintain the structural state of the material. Thus, once the MM has been switched it will retain its FRs until it is switched again. This obviously makes tunable FRs in PCMMs interesting from a ‘green technology’ perspective.

Results

Metamaterials design. The proposed MM structures consist of two metallic films (30 nm thick Au) separated by a dielectric interlayer

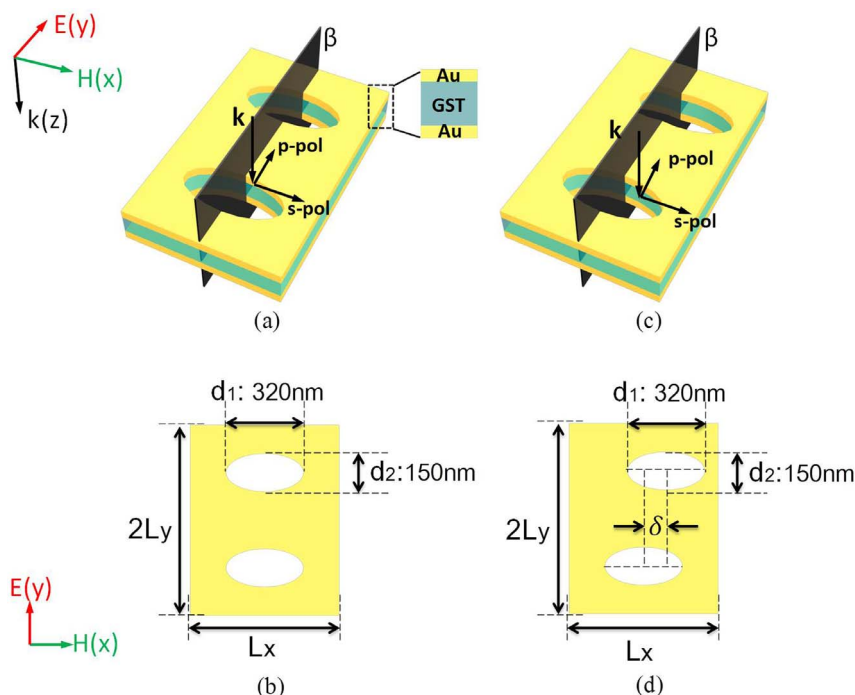


Figure 1 | (a) Schematic of the symmetric MDM–ENA ($\delta = 0$ nm) consisting of a 160 nm thick $\text{Ge}_2\text{Sb}_2\text{Te}_5$ dielectric layer between two 30 nm thick Au films perforated with a square array of elliptical holes suspended in a vacuum. (b) Illustration of the symmetric element of ENA, the lattice constants are $L_x = 500$ nm, $L_y = 400$ nm and hole diameters are $d_1 = 320$ nm, $d_2 = 150$ nm. (c) Schematic of the asymmetric MDM–ENA ($\delta = 70$ nm) consisting of a 160 nm thick $\text{Ge}_2\text{Sb}_2\text{Te}_5$ dielectric layer between two 30 nm thick Au films perforated with ENA suspended in a vacuum. (d) Illustration of the asymmetric element of ENA, the lattice constant is $L_x = 500$ nm, $L_y = 400$ nm, hole diameters are $d_1 = 320$ nm, $d_2 = 150$ nm, δ is the distance of the upper elliptical hole from the centre ($\delta = 70$ nm).

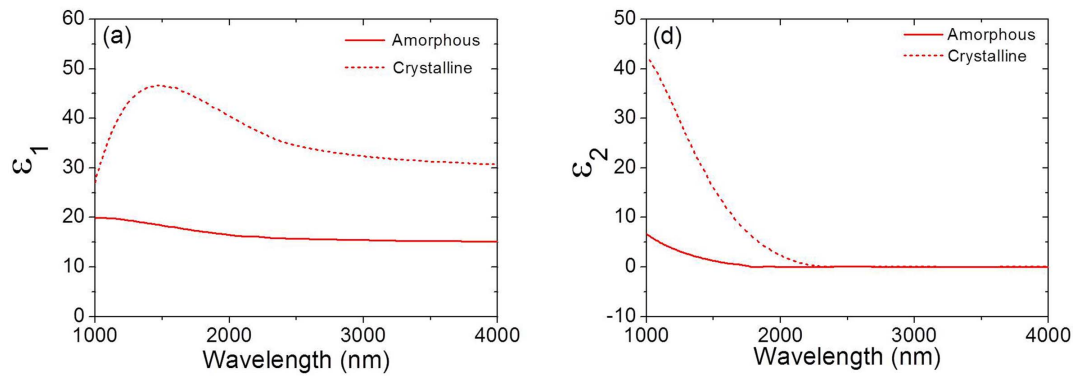


Figure 2 | Dielectric constant (a) $\epsilon_1(\omega)$ vs wavelength, (b) $\epsilon_2(\omega)$ vs wavelength for both amorphous and crystalline phases of $\text{Ge}_2\text{Sb}_2\text{Te}_5$.

(160 nm thick $\text{Ge}_2\text{Sb}_2\text{Te}_5$). Here, we simulate two sets of multilayer MMs starting from a symmetric element with both top and bottom elliptical holes exactly at the centre (see Fig. 1(a)–(b)) and then displaced the upper elliptical hole from the centre by a distance “ δ ” while keeping the lower elliptical hole fixed at the centre in order to break the symmetry (see Fig. 1(c)–(d)). For both structures, the elliptical nanohole array (ENA) (lattice constant, along the x axis is $L_x = 500$ nm, along the y axis is $L_y = 400$ nm; hole diameters, $d_1 = 320$ nm, $d_2 = 150$ nm) is perforated through the entire $\text{Au}/\text{Ge}_2\text{Sb}_2\text{Te}_5/\text{Au}$ structure, β is a cross-section plane of the structure, and the elliptical holes are periodically arranged in both the x and y directions. The Au bottom layer interacts with the upper Au layer to give rise to a closed circle of displacement current (J_D) and localize an electromagnetic (EM) field within the dielectric interlayer. The structures are considered to be suspended in a vacuum, deep etching of a silicon support substrate can be used to achieve this. Au is selected as the metal due to its stability and low ohmic loss. The

geometry of the unit cell pattern and the thickness of the sandwich layers have been chosen to allow for the impedance matching between the metamaterials and impinging plane wave⁴⁵. The simulation is performed by commercial software Comsol, which is based on the Finite Element Method (FEM). The dielectric properties of Au as given by Johnson & Christy are used⁴⁶. All the structures are excited by a source with a wavelength range from 1000 to 4000 nm, propagating along the negative z direction with the E field polarized along the small axis of the elliptical hole as shown in Fig. 1(a) and (c). The light source has a recurrence rate, $f_r = 25$ kHz and pulse duration of 0.25 ns. The light fluence illuminating the sample from a single pulse is written as⁴⁷

$$F_I(r) = \frac{2P_0}{\pi w^2 f_r} \exp\left(-\frac{2r^2}{w^2}\right) \quad (1)$$

where $P_0 = 3$ mW is the total incident power, r is the distance from the beam center, $w = 10$ μm is Gaussian beam waist. To account for

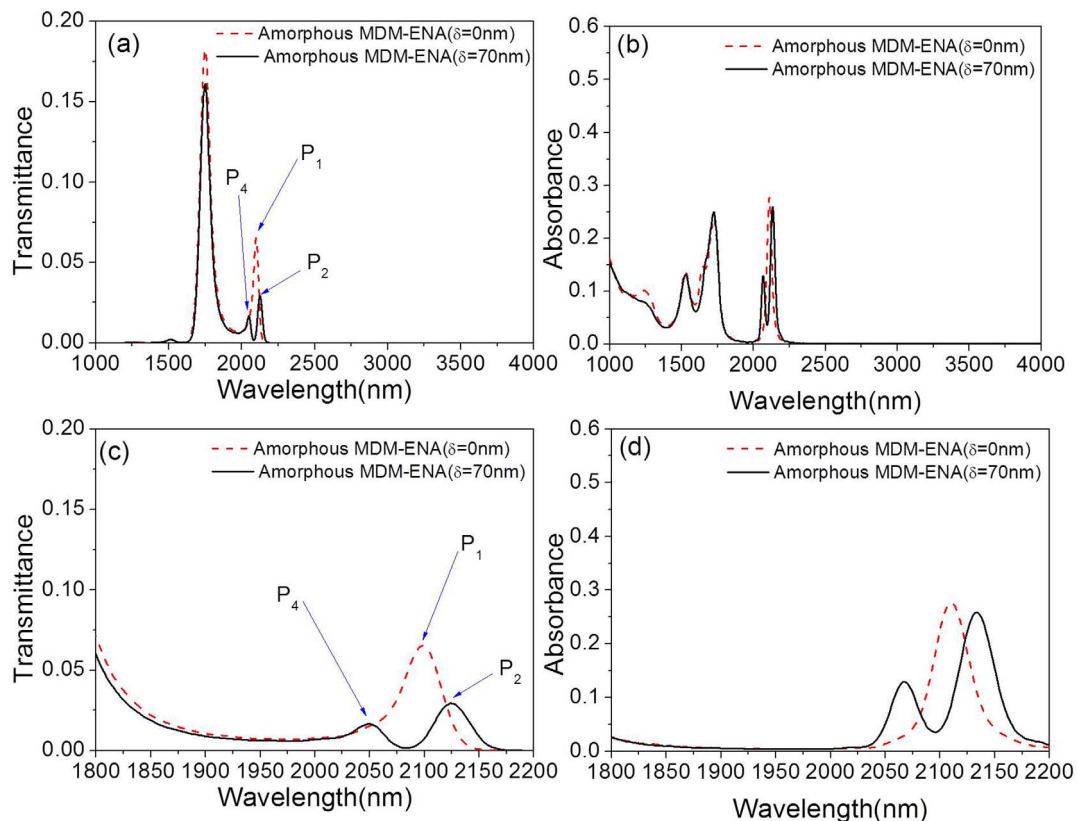


Figure 3 | The comparison of (a) the transmittance, (b) the absorbance between asymmetric MDM-ENA($\delta = 70$ nm) with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ and symmetric MDM-ENA($\delta = 0$ nm) with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$. (c) Zoom in picture of Fig. 3(a). (d) Zoom in picture of Fig. 3(b).

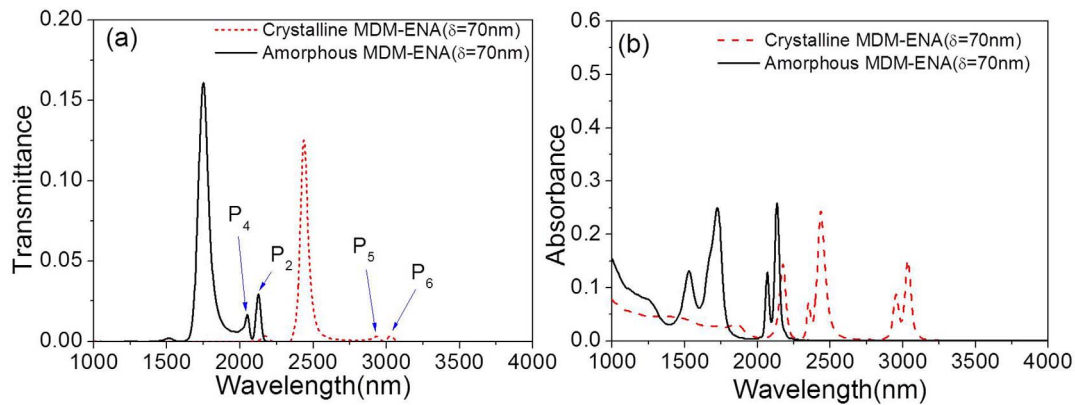


Figure 4 | The comparison of (a) the transmittance, (b) the absorbance between the asymmetric MDM-ENA($\delta = 70$ nm) with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ and crystalline $\text{Ge}_2\text{Sb}_2\text{Te}_5$.

the periodic nature of the metamaterial, the model boundary at $x = \pm \frac{L_x}{2}$ and $y = \pm L_y$ is set to condition of perfect magnetic conductor and perfect electric conductor, respectively for normal incidence. Scattering boundary conditions are applied in the z direction. Adaptive meshes are generated by a built-in auto meshing algorithm of Comsol. The frequency dependent reflection $r(\omega)$ and transmission $t(\omega)$ are obtained from the S-parameters in the simulation package, and the absorbance is calculated as $1 - |r(\omega)|^2 - |t(\omega)|^2$ ⁴⁸.

Optical properties of $\text{Ge}_2\text{Sb}_2\text{Te}_5$. The real, $\epsilon_1(\omega)$ and imaginary, $\epsilon_2(\omega)$ parts of the dielectric function of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ in different phases are obtained from the experimental data in⁴², and for the mid-infrared (M-IR) spectral range the dielectric function is shown in Fig. 2; as can be seen there is a very large change in the dielectric function's real component. The dielectric constant of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ is very dispersive and has a non-negligible imaginary part. It also changes considerably during the reversible structural transformation from amorphous to crystalline. Different PCMs can display similar behavior in other parts of the spectrum. It should be mentioned that the reversible phase transition in $\text{Ge}_2\text{Sb}_2\text{Te}_5$ is highly repeatable and more than a billion cycles have been experimentally demonstrated in data storage devices⁴⁹.

Tunability of double Fano resonance in the phase-change metamaterial. Figure 3 presents transmittance and absorbance for the structures of symmetric MDM-ENA($\delta = 0$ nm) and asymmetric MDM-ENA($\delta = 70$ nm) at normal incidence, where the dielectric layer is amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$. Here, we show that the symmetric MDM-ENA exhibits a single FR (denoted as P1 mode) with

resonance wavelength centered at about 2098 nm (898.5 THz) (red dash curve in Fig. 3(a) with zoom-in in Fig. 3(c)). Spectral width of the single FR is 13.4 THz, leading to a quality factor(Q) of 67, where Q factor is defined by the resonant frequency over the width of the resonance⁵⁰. It can be inferred that this single FR must be excited by the interaction between the electric resonance and the magnetic resonance in the structure. By displacing the upper elliptical hole from its centre by a distance $\delta = 70$ nm, the single FR is split in two distinct resonant peaks at 2124 nm (887.5 THz) (marked as P2 mode) and 2050 nm (marked as P4 mode) by an asymmetric resonant dip. Namely, the single magnetic resonance at the wavelength of 2098 nm would be divided into two magnetic resonances at different resonance wavelengths of 2050 nm and 2124 nm by breaking the structural symmetry. Therefore, the double FR in the transmittance spectrum (black solid curve in Fig. 3(a)) is attributed to the destructive interference of the electric resonance (bright modes) with the magnetic resonances (dark modes) at different resonant wavelengths. In particular, a narrow transparency sub-band resonance window with a spectral width of 10.3 THz is achieved at P2 mode for the asymmetric MDM-ENA. It gives rise to a higher Q factor of 86 than the Q factor of 67 in the symmetric structure. This shows that asymmetric elements can significantly improve the FR in 3D multilayer MMs. In Figs. 3(b) and 3(d), one can observe that the absorbance in the asymmetric MDM-ENA is also split accordingly, whereas possessing a similar maximum value with the symmetric MDM-ENA.

Figure 4 shows the evolution of the double FRs in both the transmittance and absorbance spectrums of the asymmetric MDM-ENA($\delta = 70$ nm) for M-IR radiation at normal incidence, in the amorphous and metastable cubic crystalline states. With the phase change of $\text{Ge}_2\text{Sb}_2\text{Te}_5$, the double FR at 2124 nm (P2 mode) and

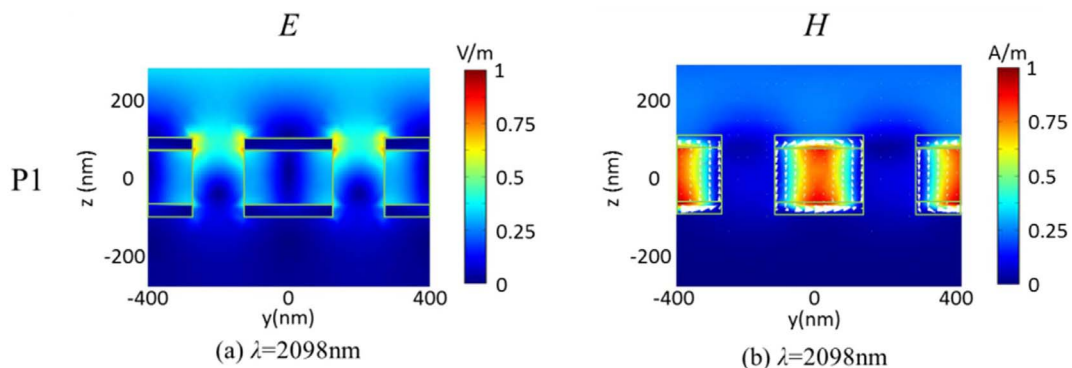


Figure 5 | 3D-FEM simulation of (a) total electric field intensity distribution, (b) total magnetic field intensity distribution and J_D distribution along β plane for the amorphous symmetric MDM-ENA($\delta = 0$ nm) at P1 mode($\lambda = 2098$ nm).

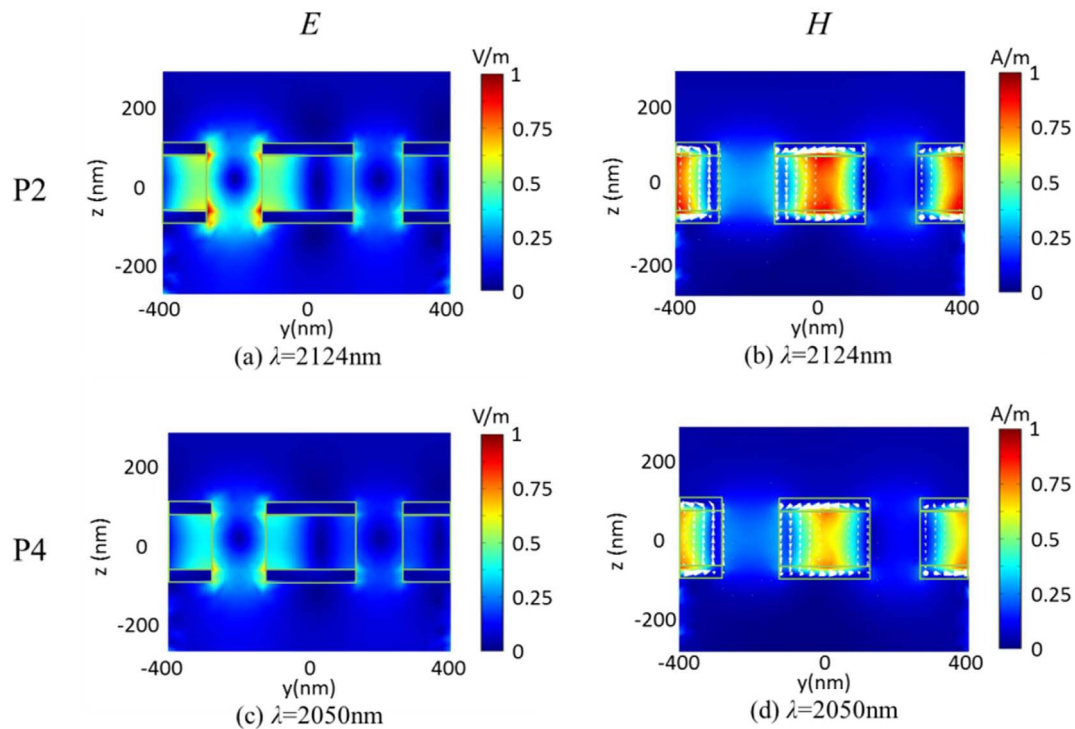


Figure 6 | 3D-FEM simulation of (a) total electric field intensity distribution, (b) total magnetic field intensity distribution and J_D distribution along β plane, at P2 mode($\lambda = 2124$ nm); (c) total electric field intensity distribution, (d) total magnetic field intensity distribution and J_D distribution along β plane, at P4 mode($\lambda = 2050$ nm) for the amorphous asymmetric MDM-ENA($\delta = 70$ nm).

2050 nm(P4 mode) for the amorphous state shifts to a double FR at 2932 nm (P5 mode) and 3028 nm(P6 mode) for the crystalline state. Figure 4(b) presents the trapped plasmon mode absorbance lines of the structure for different states of $\text{Ge}_2\text{Sb}_2\text{Te}_5$. The $\text{Ge}_2\text{Sb}_2\text{Te}_5$ variable dielectric function gives rise to a concomitant

tunability in the metamaterial absorbance. We also find that the peaks of both transmittance and absorbance drop down when $\text{Ge}_2\text{Sb}_2\text{Te}_5$ changes from the amorphous to crystalline in addition to a shift, owing to a reduction in the impedance matching of the structure to vacuum.

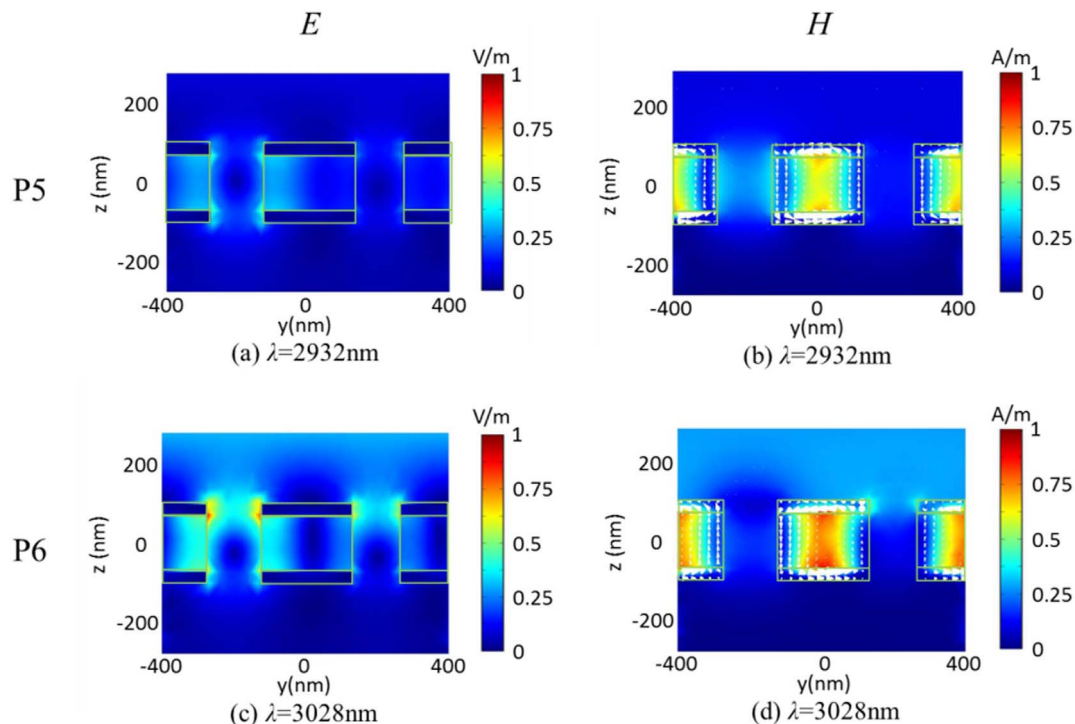


Figure 7 | 3D-FEM simulation of (a) total electric field intensity distribution, (b) total magnetic field intensity distribution and J_D distribution along β plane, at P5 mode($\lambda = 2932$ nm); (c) total electric field intensity distribution, (d) total magnetic field intensity distribution and J_D distribution along β plane, at P6 mode($\lambda = 3028$ nm) for the crystalline asymmetric MDM-ENA($\delta = 70$ nm).

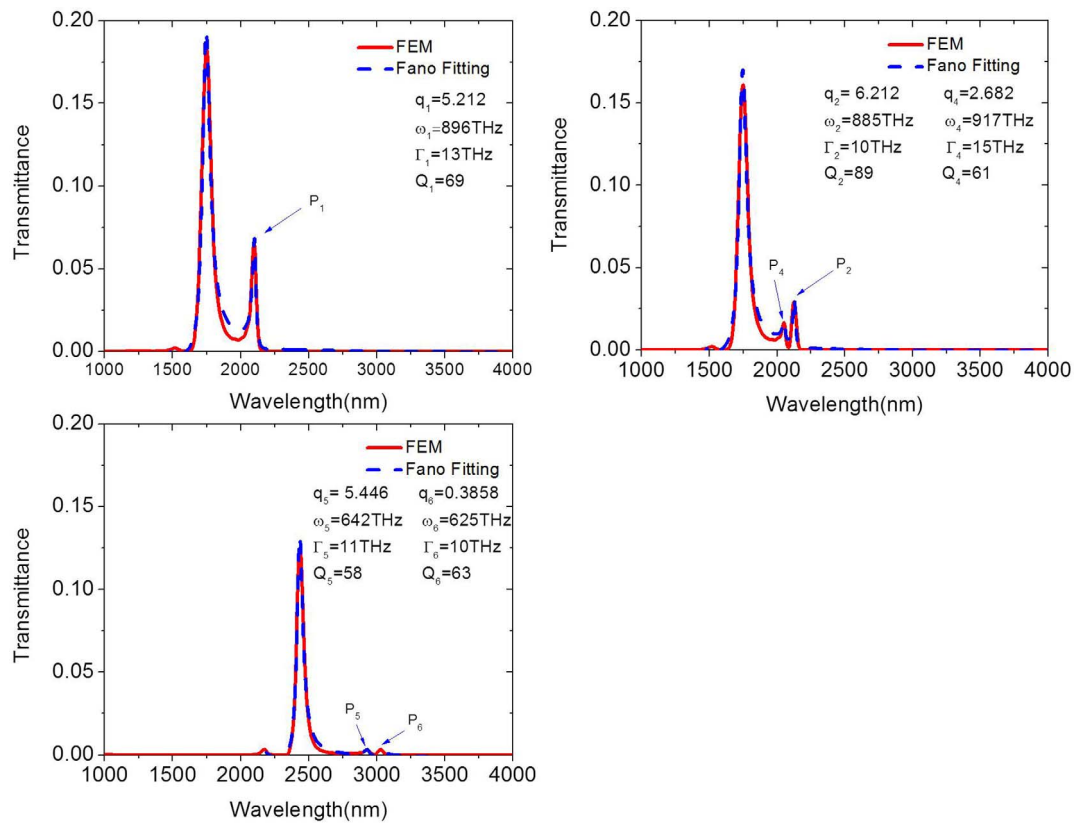


Figure 8 | (a) 3D-FEM simulation of the transmittance spectrums (red solid) of the symmetric MDM-ENA with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ and best-fits to the FR lineshape Eq.(1) (blue dash); (b) 3D-FEM simulation of the transmittance spectrums (red solid) of the asymmetric MDM-ENA with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ and best-fits to the FR lineshape Eq.(1) (blue dash); (c) 3D-FEM simulation of the transmittance spectrums (red solid) of the asymmetric MDM-ENA with crystalline $\text{Ge}_2\text{Sb}_2\text{Te}_5$ and best-fits to the FR lineshape Eq.(1) (blue dash).

Figure 5 shows total electric field intensity distributions $E = \sqrt{|E_x|^2 + |E_y|^2 + |E_z|^2}$, total magnetic field intensity distributions $H = \sqrt{|H_x|^2 + |H_y|^2 + |H_z|^2}$ and J_D along the β plane at the resonant wavelengths of 2098 nm (P1 mode) for the symmetric MDM-ENA ($\delta = 0$ nm) with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$. Here, the arrows represent J_D whereas the color represents the magnitude of the electromagnetic field intensities. In Fig. 5(b), the formation of the closed J_D loops are observed to support a magnetic resonance at which light is trapped and strongly absorbed, thus the H field can be efficiently confined between two Au layers. However, Fig. 5(a) shows a low concentration of the E field intensity within the dielectric interlayer and elliptical apertures, indicating a weak electric resonance.

Figure 6 demonstrates E, H and J_D along the β plane at the resonant wavelengths of 2124 nm (P2 mode) and 2050 nm (P4 mode) for the asymmetric MDM-ENA ($\delta = 70$ nm) with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$. Figure 6(b) shows that H field intensity distributions are still confined well within the dielectric layer at P2 mode.

Whereas, Figure 6 (d) shows that H field intensity at P4 mode is weaker hence leading to a smaller magnetic resonance dipole than both P1 in Fig. 5(b) and P2 mode in Fig. 6 (b). Figure 6(a) and (c) show the E field intensity distributions at P2 and P4 mode, respectively. As can be seen, the E field intensity distributions at P2 mode in Fig. 6(a) are more efficiently concentrated thus giving rise to much stronger electric resonances than P1 in Fig. 5(a) and P4 mode in Fig. 6(c). Therefore, for the asymmetric MDM-ENA, the interaction of the stronger electric resonance and magnetic resonance leads to the narrower FR at P2 mode compared with P1 mode in the symmetric MDM-ENA.

In Fig. 7(a)–(d), we present the distributions of E, H and J_D associated with P5 ($\lambda = 2932$ nm) and P6 ($\lambda = 3028$ nm) modes for the asymmetric MDM-ENA ($\delta = 70$ nm) with crystalline $\text{Ge}_2\text{Sb}_2\text{Te}_5$. It demonstrates that the closed J_D loops are still maintained at two resonance modes. It provides two magnetic dipolar moments thus leading to double resonant peaks. However, the double FRs is broadened owing to the inference between the weak magnetic resonances

Table 1 | Parameters used in the FR lineshape to fit the transmittances of the MDM-ENA

Pi	q_i	ω_i (THz)	Γ_i (THz)	Q_i
P1	5.212	896	13	69
P2	6.212	885	10	89
P4	2.682	917	15	61
P5	5.446	642	11	58
P6	0.3858	625	10	63

Table 2 | Material thermal properties used in the Heat transfer model

	Special heat capacity Cs (J/(kg·K))	Density ρ (kg/m ³)	thermal conductivity k(W/(m·K))
gold	129 ⁵⁶	19300 ⁵⁶	317(bulk) ⁵⁶ , 91 (thickness = 30 nm) ⁵⁶ ,
$\text{Ge}_2\text{Sb}_2\text{Te}_5$	220 ⁵⁵	6150 ⁵⁵	Temperature dependence ⁵⁵
air	1 ⁴⁷	$\frac{353[K]}{T}$ ⁴⁷	0.03 ⁴⁷

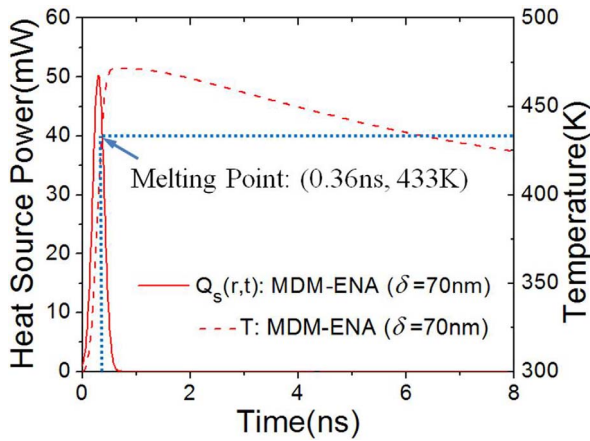


Figure 9 | 3D-FEM simulation of heat power irradiating on an amorphous asymmetric MDM-ENA located at the beam center, where the solid red line presents the heat power irradiating on the structures under normal incident intensity of $9.6 \mu\text{W}/\mu\text{m}^2$, the dash red line is the temperature of the amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ layer during one pulse.

and electric resonances, resulting from the low intensities distributions of H and E fields in the structure.

Fano formula fitting to the transmittance spectrums simulated through FEM method. In Fig. 8, we fit the transmittance curves of both the symmetric MDM-ENA ($\delta = 0 \text{ nm}$) and asymmetric MDM-ENA ($\delta = 70 \text{ nm}$) using the FR lineshape in Eq.(2)^{51–53}:

$$T \propto \sum_i \frac{(\varepsilon_i + q_i)^2}{1 + \varepsilon_i^2}, (i = 1, 2, \dots) \quad (2)$$

where $\varepsilon_i = \frac{2(\omega - \omega_i)}{\Gamma_i}$, particularly, q_i are the phenomenological shape parameters (so-called asymmetry parameters), ω_i are the resonance frequencies and Γ_i are the widths of the autoionized states (the resonance linewidth at half maximum, FWHM) of the resonant modes $\text{Pi}^{53,54}$. With the parameters shown in Table 1, the calculated transmittances based on Eq.(1) approximately reproduce our FEM simulation data for the symmetric MDM-ENA ($\delta = 0 \text{ nm}$) with amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$, asymmetric MDM-ENA ($\delta = 70 \text{ nm}$) with amorphous and crystalline $\text{Ge}_2\text{Sb}_2\text{Te}_5$, respectively.

It is found that the Fano formula mostly matches the FEM simulation data around the resonant modes. Q factor is defined by the resonant frequency over the width of the autoionized state^{15,51}, where the resonant frequency and the width of the autoionized state are obtained from the Fano formula fitting transmittance curves shown in Fig. 8.

Photothermal effect of the phase-change metamaterial. Since the reversible amorphous - crystalline phase transition of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ can be induced through optical heating, it is essential to understand the heat induced switching behavior of the asymmetric MDM-ENA. To show this, a heat transfer model is used to investigate the temporal variation of temperature of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ layer at normal incidence using the Finite Element Method (FEM) solver within COMSOL. The material thermal properties used for the simulation are summarized in Table 2. The thermal conductivity of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ changes with the temperature is obtained from the experiment data in⁵⁵.

In this simulation, the thermal energy absorbed by one unit cell is defined as⁴⁷

$$E_{th}(r) = R_a \times L_x \times L_y \times F_I(r) \quad (3)$$

where $L_x = 500 \text{ nm}$ is the lattice constant of the structure along the x axis; $L_y = 400 \text{ nm}$ is the lattice constant of the structure along the y axis; $R_a = 0.0438$ is the absorption coefficient of the structure, derived from the overlap integral between the light source power density spectrum and the absorbance spectrum, shown in Fig. 4(b). The heat source power is then described by a Gaussian pulse function

$$Q_s(r, t) = E_{th}(r) \frac{1}{\sqrt{\pi}\tau} \exp\left(-\frac{(t - t_0)^2}{\tau^2}\right) \quad (4)$$

where $\tau = 0.15 \text{ ns}$ is the time constant of the light pulse, $t_0 = 0.3 \text{ ns}$ is the time delay of the pulse peak. Figure 9 shows $Q_s(r, t)$ and the temperature of the amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ layer, where the structure is located at the center of Gaussian beam. The numerical simulation shows that the temperature within the amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ dielectric layer is a function of the incident radiation flux and can exceed the amorphous to crystalline phase transition temperature of 433 K after 0.36 ns and a maximum of 472 K after 0.59 ns under a threshold incident flux of $9.6 \mu\text{W}/\mu\text{m}^2$. The temperature starts dropping after 0.59 ns before the next pulse arrives, due to heat dissipation to the surroundings.

The temperature distributions of the structure at 0.36 ns and 0.59 ns along the plane β are shown in Fig. 10(a) and (b) respectively. One can observe that the temperature within the amorphous $\text{Ge}_2\text{Sb}_2\text{Te}_5$ layer is uniform, and the dominant temperature gradient is towards the top and bottom Au films.

Discussion

In conclusion, we have numerically demonstrated a low-power and rapid all-optical tunable double FR in an asymmetric fishnet MMs based on a $\text{Au}/\text{Ge}_2\text{Sb}_2\text{Te}_5/\text{Au}$ multilayer. By breaking the symmetry in the ENA element, the single FR from the symmetric MDM-ENA splits into two resonant peaks, resulting in a double FR. This interference leads to a sharp transparency peak with higher Q factor. A

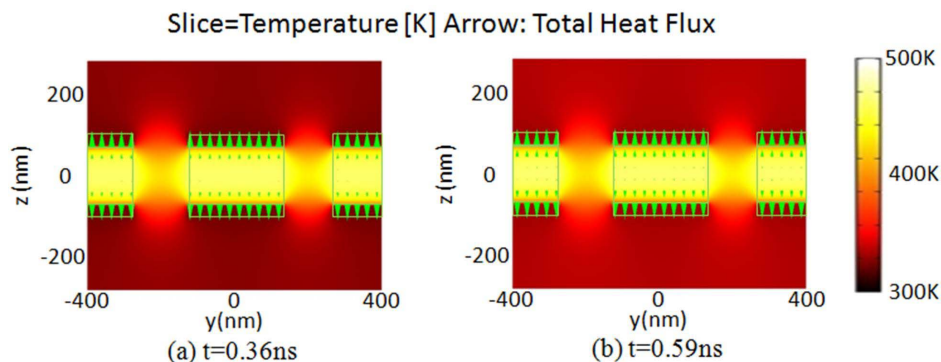


Figure 10 | The temperature distribution of the unit cell of the amorphous asymmetric MDM-ENA along the β plane at (a) 0.36 ns and (b) 0.59 ns, where the color image indicates the temperature distribution and the arrows indicate the heat flux.



wide wavelength shift of 42% for double FRs is observed in the M-IR region. This tunable effect is due to the phase transition from amorphous to crystalline. A heat transfer model is built to resolve the transient temperature variation in the structure during a photo-thermal process. Our model predicts that a fast phase transition time of 0.36 ns can be reached under a weak pump intensity of $9.6 \mu\text{W}/\mu\text{m}^2$. This work introduces a new approach for the study of ultralow-power and ultrafast photonic nanodevices. It may open up avenues for designing active multiple-wavelength slow light devices, plasmonics switching and sensors.

Methods

For the numerical calculations, we used the Finite Element Method by means of COMSOL Multiphysics. The unit cell of the phase-change metamaterial is shown in Fig. 1. The incident light is normal to the x - y plane with E field polarized in y direction. To account for the periodic nature of the metamaterial, the model boundary at $x = \pm \frac{L_x}{2}$ and $y = \pm \frac{L_y}{2}$ is set to condition of perfect magnetic conductor and perfect electric conductor, respectively for normal incidence. Scattering boundary conditions are applied in the z direction. The optical constant of gold is taken from the data of Johnson and Christy⁴⁶. The dielectric function of $\text{Ge}_2\text{Sb}_2\text{Te}_5$ in different phases are obtained from the experimental data in⁴².

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Author contributions

T.C., M.J.C. and R.E.S. conceived the idea of using phase-changed material for tuning the Fano resonance in the metamaterials. T.C. designed the metamaterial and performed numerical analysis. C.W.W. and L.Z. carried out the simulations and prepared the figures. T.C. supervised the research. T.C. and M.J.C. co-wrote the manuscript.

Additional information

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